

# Quantifying the oxygen stoichiometry of Pr-doped ceria through X-ray diffraction

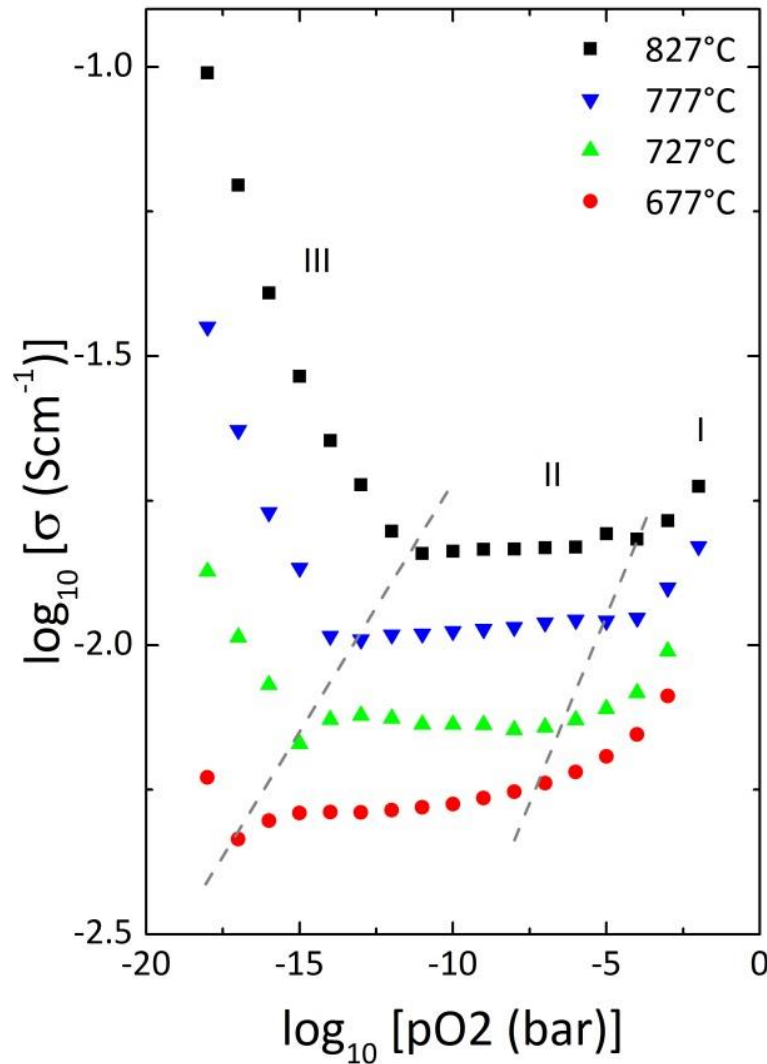
**Christian Lenser<sup>1</sup>, Felix Gunkel<sup>2</sup>, Yoo Jung Sohn<sup>1</sup> and Norbert H. Menzler<sup>1</sup>**

<sup>1</sup>: Forschungszentrum Jülich GmbH, Institute of Energy and Climate Research - Materials Synthesis and Processing (IEK-1), 52428 Jülich, Germany

<sup>2</sup>: Institute for Electronic Materials (IWE2), RWTH Aachen University, 52074 Aachen, Germany

- **Background and Motivation**
- HT-XRD: oxygen stoichiometry and microstrain
- Impact of conductivity on cathode performance
- Summary

# High Temperature Equilibrium Conductivity



HTEC of dense 20PCO ceramics

Region I:

MIEC ( $[V_O^{**}]$ ,  $[Pr'_{Ce}]$ )

Region II:

ionic plateau ( $2[V_O^{**}] \approx [Pr'_{Ce}] = [Pr_{Ce}]$ )

Region III:

Reduction of Ce (MIEC)

DC conductivity as expected for PCO<sup>1,2</sup>

<sup>1</sup>: Bishop *et al.*, J. Mater. Res., Vol. 27, No. 15, Aug 14, 2012

<sup>2</sup>: Bishop *et al.*, Phys. Chem. Chem. Phys., 2011, **13**, 10165–10173

# PCO as a cathode material

Figure of merit:  $L_C = D^*/k^a$

$D^*$ : oxygen self-diffusion coefficient

$k^a$ : oxygen surface exchange coefficient

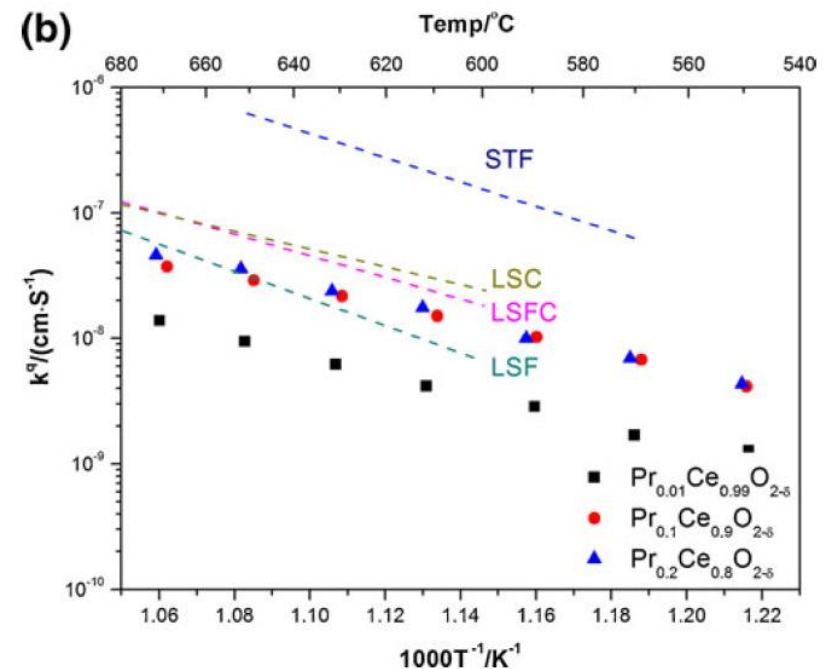
Model system suggests high  $k^a$  <sup>1</sup>

Investigations on:

PCO impregnated LSM-YSZ<sup>2</sup>

PCO interlayer for  $\text{LaNi}_{0.6}\text{Fe}_{0.4}\text{O}_3$  <sup>3</sup>

PCO as an active material<sup>4</sup>



<sup>1</sup>: Chen et al., J. Electroceram. (2012) 28:62-69

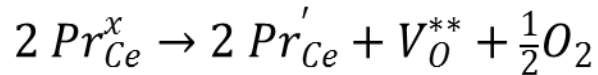
<sup>2</sup>: Y. Ren, et al., Journal of Materials Chemistry **22** (2012) (48) 25042.

<sup>3</sup>: R. Chiba, et al., Electrochemical and Solid-State Letters **12** (2009) (5) B69.

<sup>4</sup>: R. Chiba, et al., Solid State Ionics **197** (2011) (1) 42.

# Conductivity of $\text{Pr}_{1-x}\text{Ce}_x\text{O}_{2-\delta}$

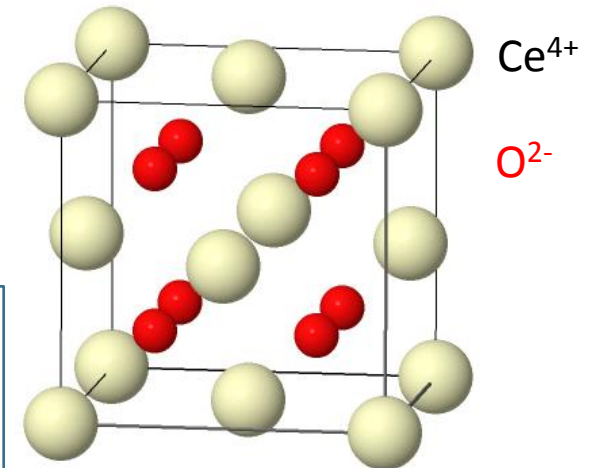
## Reduction



## Conductivity

$$\sigma_{\text{tot}} = \sum_i \sigma_i = \sum_i n_i q_i \mu_i$$

$$\sigma_{\text{tot}} = \sigma_{\text{Pr}} + \sigma_{\text{ion}} = [\text{Pr}_{\text{Ce}}'] * |e| * \mu_{\text{Pr}} + [V_{\text{O}}^{**}] * 2|e| * \mu_V$$



## Charge neutrality

$$2[V_{\text{O}}^{**}] \approx [\text{Pr}_{\text{Ce}}'] + n$$

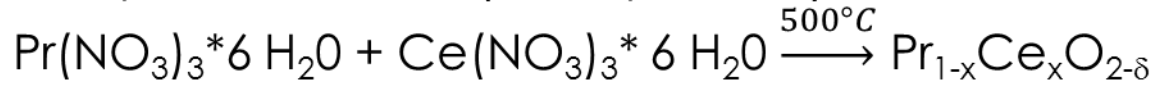
$$\begin{array}{l} \text{High } p\text{O}_2 \\ n \ll [\text{Pr}_{\text{Ce}}'] \end{array}$$

$$\mu_i = \frac{\mu_{0,i}}{T} \exp\left(-\frac{H_{m,i}}{kT}\right)$$

$$\sigma = [\text{Pr}_{\text{Ce}}'] * e \left[ \frac{\mu_{0,\text{Pr}}[\text{Pr}_{\text{Ce}}^x]}{T[\text{Pr}_{\text{Ce}}]} \exp\left(-\frac{H_{m,\text{Pr}}}{kT}\right) + \frac{\mu_{0,\text{V}}}{T} \exp\left(-\frac{H_{m,\text{V}}}{kT}\right) \right]$$

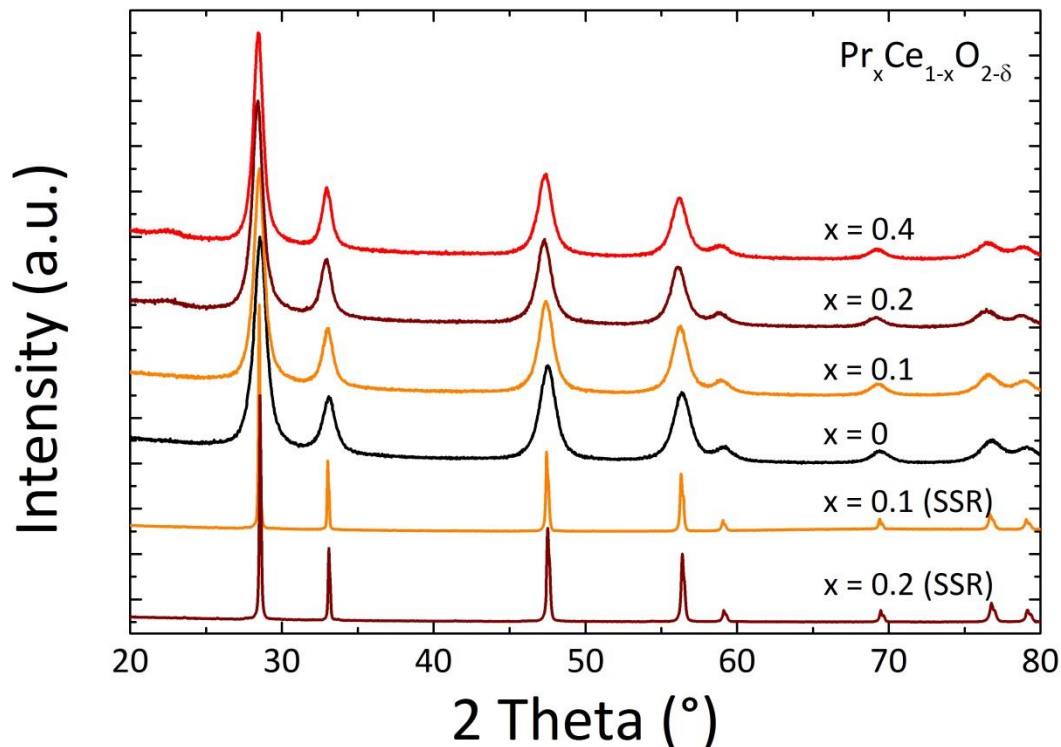
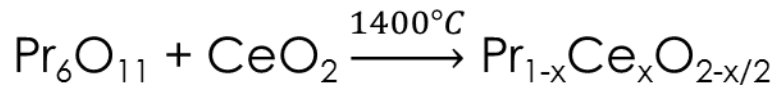
- Background and Motivation
- **HT-XRD: oxygen stoichiometry and microstrain**
- Impact of conductivity on cathode performance
- Summary

„Simplified Pechini“ (Nanopowder)



Dissolution, gelling and combustion of nitrates

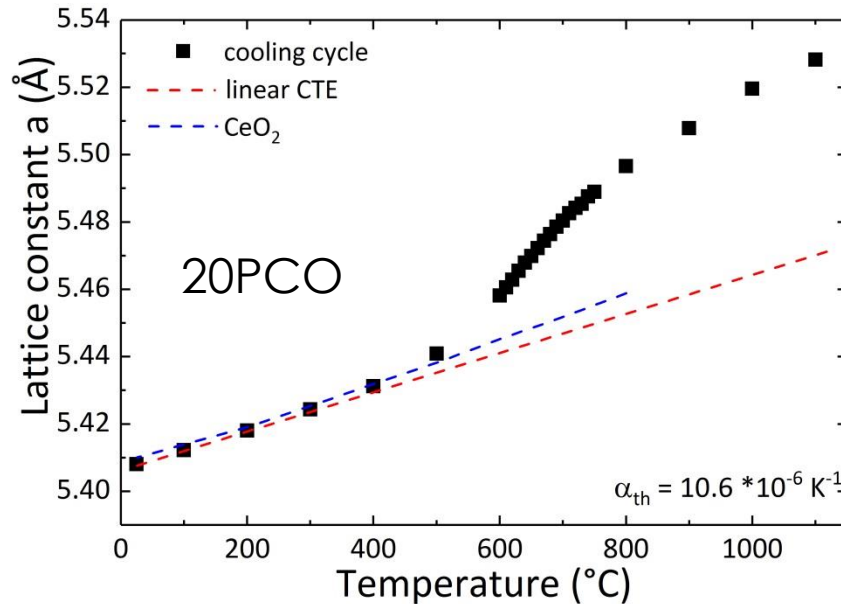
## Solid State Route



Line broadening:

- Crystallite size
- Microstrain ( $\Delta d_{hkl}$ )

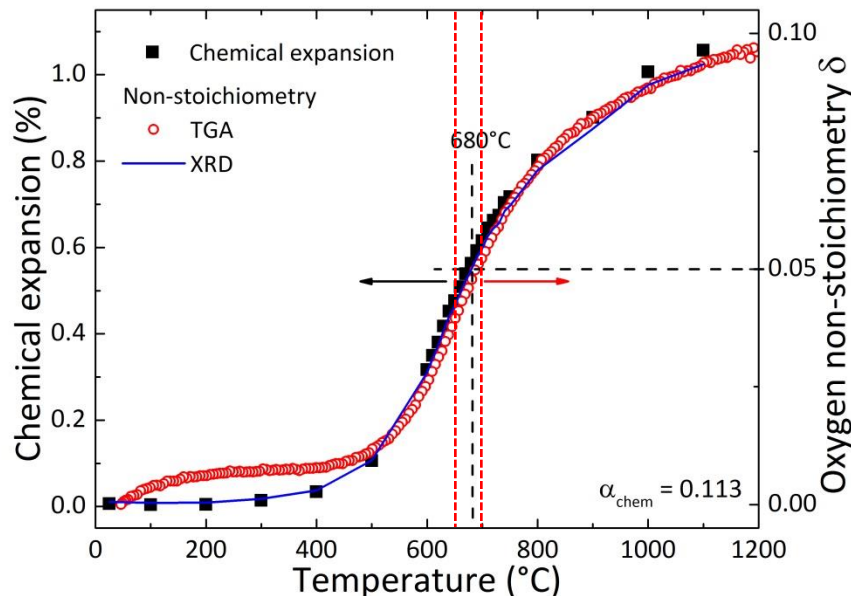
# Chemical expansion in air



- Chemical expansion in air

- $\alpha_{tot} = \alpha_{thermal} + \alpha_{chem}$

- $\alpha_{thermal} = \frac{1}{a} \frac{da}{dT} = 10.6 \cdot 10^{-6} \text{ K}^{-1}$

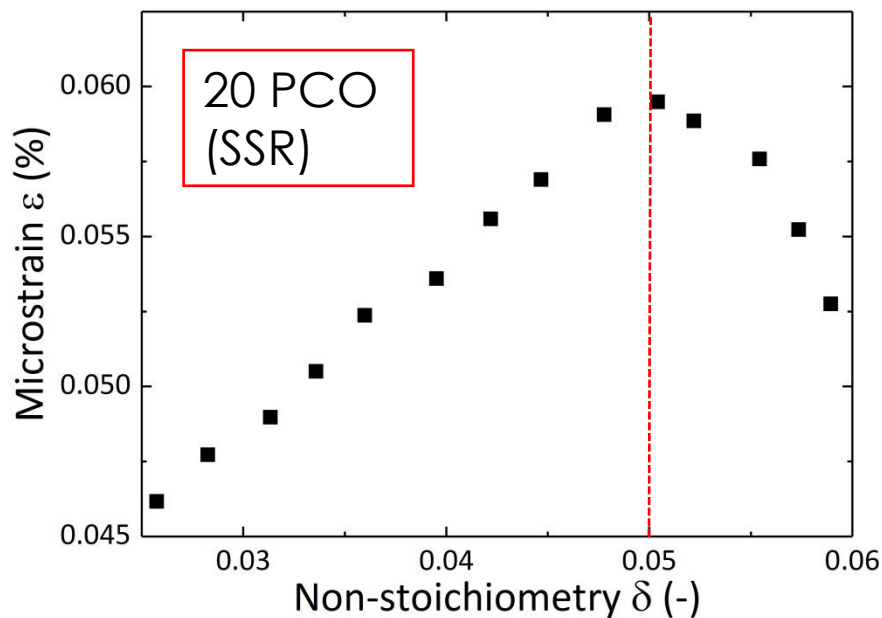
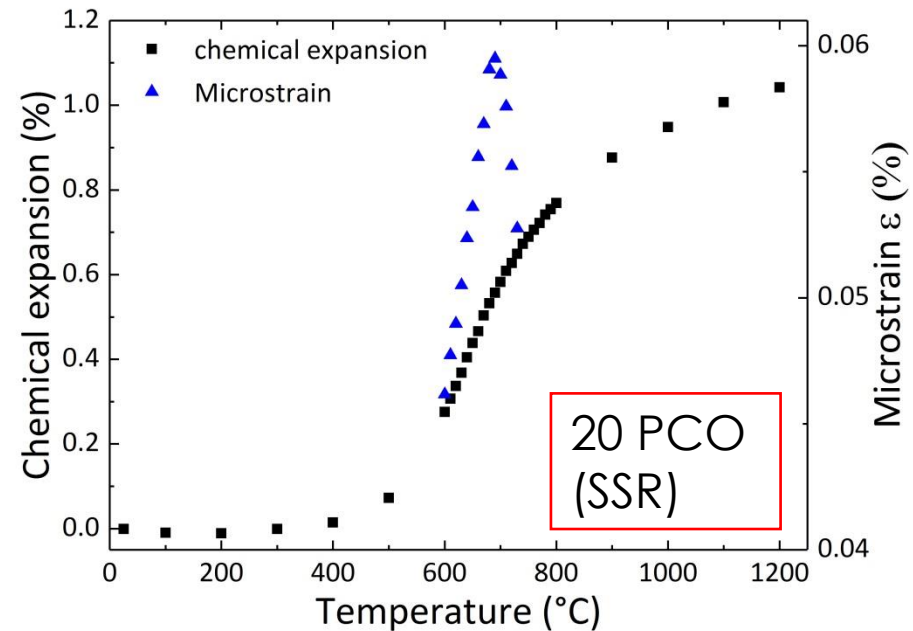
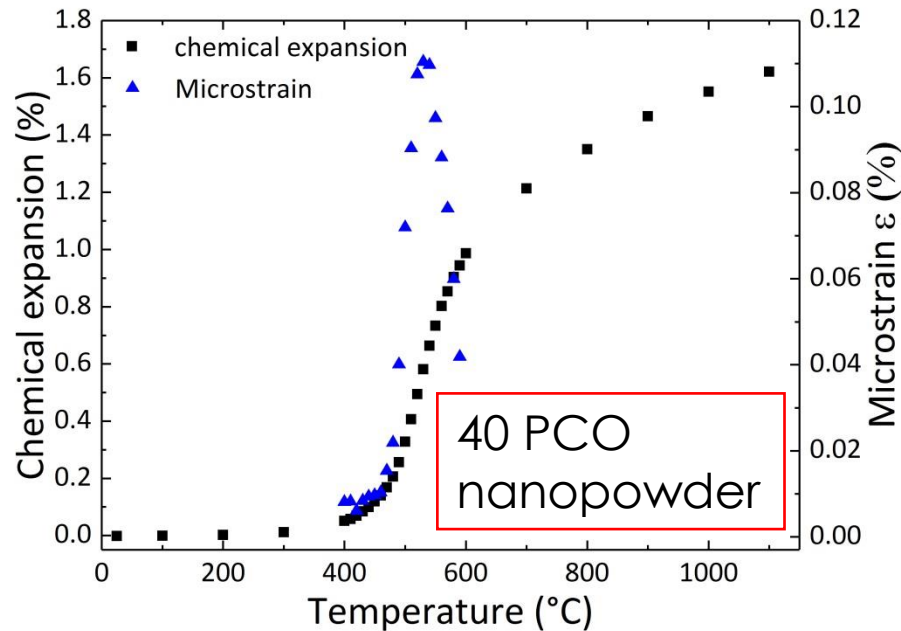


- $\alpha_{chem} = \frac{da}{d\delta} = 0.113$

- $\delta = 0.05$  at  $\sim 680^\circ\text{C}$



# Microstrain analysis



- Microstrain:  $\Delta d_{hkl}$

- Background and Motivation
- HT-XRD: oxygen stoichiometry and microstrain
- **Impact of conductivity on cathode performance**
- Summary

# Conductivity of 20PCO in air

$$\sigma = [Pr'_{Ce}] * e \left[ \frac{\mu_{0,Pr} [Pr^x_{Ce}]}{T [Pr_{Ce}]} \exp\left(-\frac{H_{m,Pr}}{kT}\right) + \frac{\mu_{0,V}}{T} \exp\left(-\frac{H_{m,V}}{kT}\right) \right]$$

Ionic plateau ( $pO_2 = 10^{-12}$  bar)

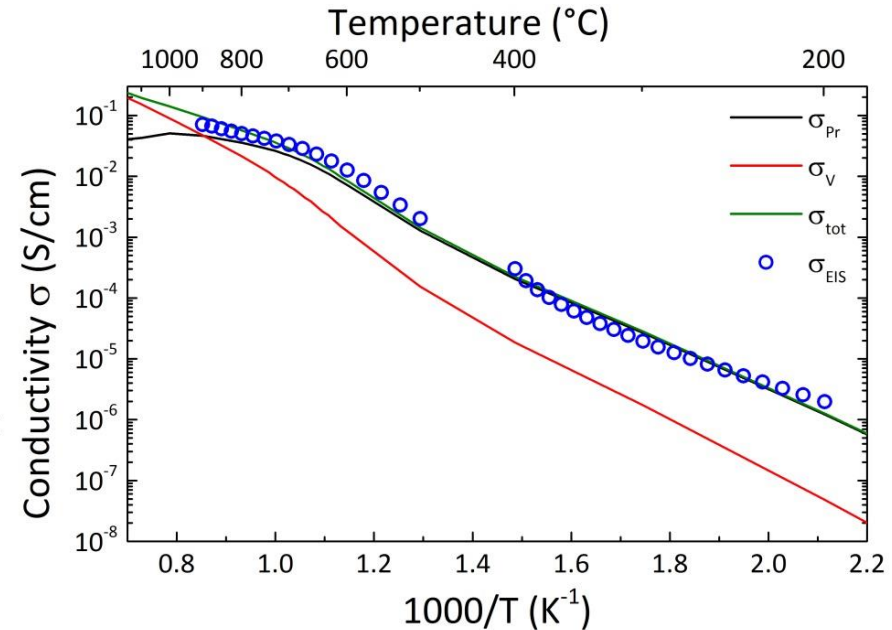
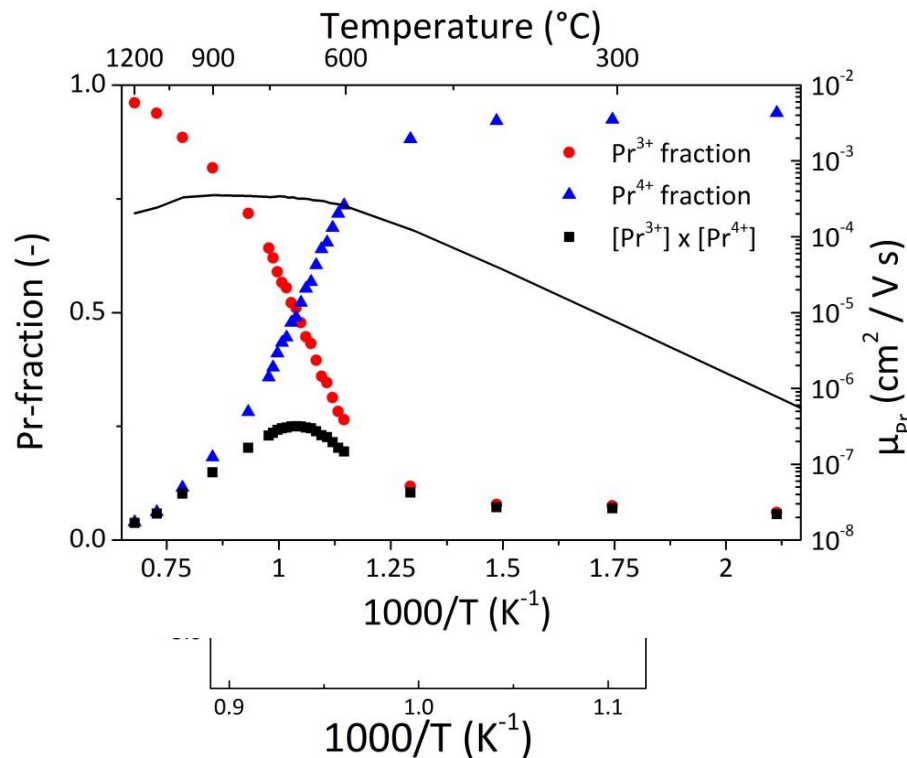
$$H_{m,V} = 0.83 \text{ eV}$$

$$\mu_{0,V} = 327 \text{ cm}^2/\text{Vs}$$

$$\sigma_{Pr} = \sigma_{tot} - \sigma_{ion}$$

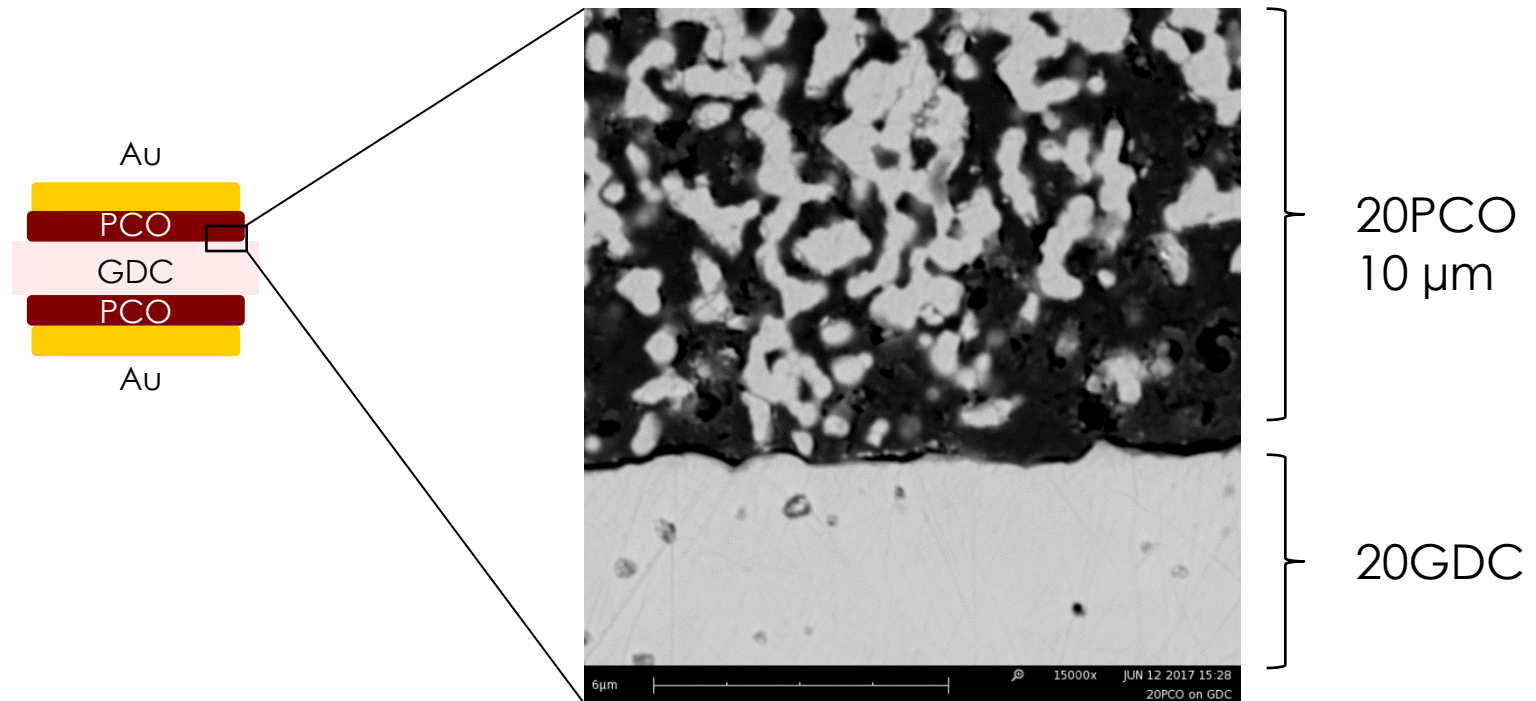
$$H_{m,Pr} = 0.72 \text{ eV}$$

$$\mu_{0,Pr} = 589 \text{ cm}^2/\text{Vs}$$



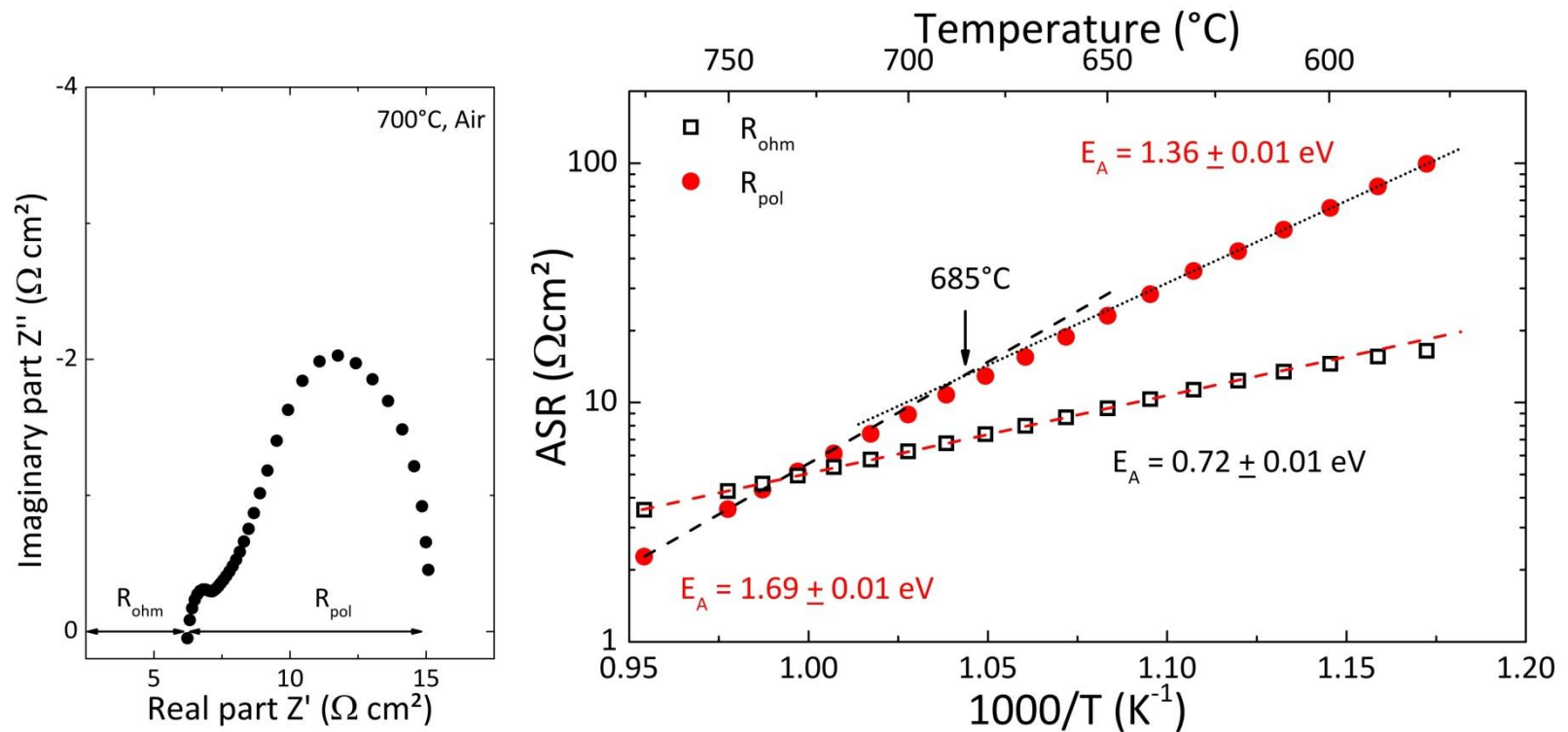
# Sample preparation

Screen printed samples on 1 mm 20GDC pellet



Cathode sintering at 1300°C

# Electrochemical impedance spectroscopy



## Assumptions:

$R_{\text{ohm}}$  = ionic resistance of electrolyte

$R_{\text{pol}}$  = oxygen surface exchange resistance of electrode

- Conductivity of 20PCO sensitive to Pr-valence state
- Mixed valence of Pr induces microstrain – maximum at  $[\text{Pr}^{3+}] = [\text{Pr}^{4+}]$
- Polaron mobility has maximum at  $\delta = 0.05$  – 680°C for 20PCO in air
- Polaron mobility influences polarization resistance  $R_{\text{pol}}$
- Activation energy of 20PCO cathodes increases with T
- Chemical expansion a problem for processing



**Thank you for your attention!**

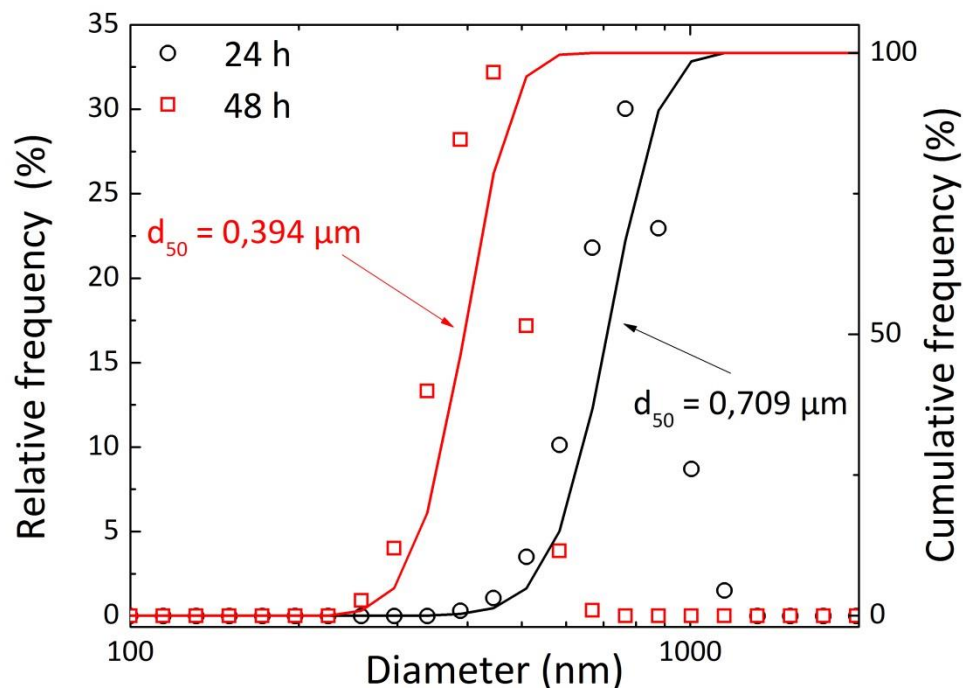




# Preparation of screen printing pastes

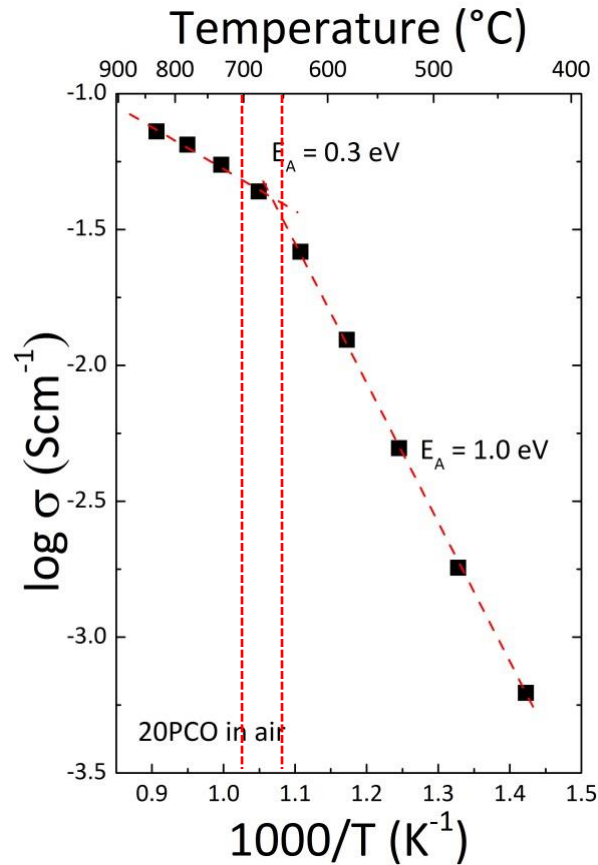
Grain size is very important for properties of ceramic

Control of particle size via ball-milling for solid state powders



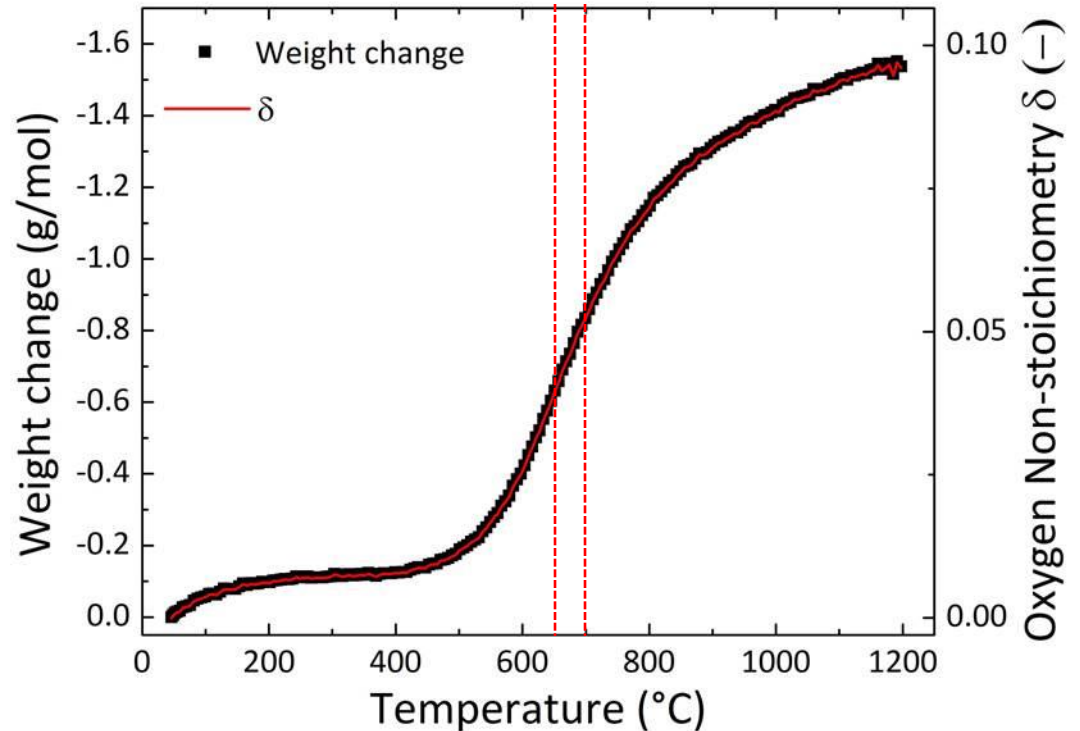
Screen printing pastes: powders in terpineol mixing the pre-suspension with ethylcellulose solved in terpineol. Layers were printed on  $150 \mu\text{m}$  8YSZ foils (Kerafol), using a circular geometry with  $\varnothing = 12 \text{ mm}$  for PCO and  $\varnothing = 10 \text{ mm}$  for LSCF. Symmetric and asymmetric cells were prepared for EIS and microscopy.

## 4-point conductivity



- Change in  $E_A$

## Thermogravimetry



- Temperature interval close to  $\delta = 0.05$